

# Surfactant-assisted electrodeposition and improved electrochemical capacitance of silver-doped manganese oxide pseudocapacitor electrodes

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**Abstract** Ag-doped MnO<sub>2</sub> pseudocapacitor electrodes with dendrite and foam-like structures were successfully produced for the first time using an electrodeposition method employing structure-directing agents, i.e., sodium dodecyl sulfate (SDS) and cetyltrimethylammonium bromide (CTAB) acting through micelle formation at solid–liquid interfaces. Doping silver with MnO<sub>2</sub> enhanced their electronic conductance. Controlling pseudocapacitor electrode morphologies with surfactants accelerated ion transport. The specific capacitance values of the Ag-doped MnO<sub>2</sub> films produced with SDS and CTAB, measured in 0.5 M Na<sub>2</sub>SO<sub>4</sub>

at a scan rate of 5 mV s<sup>-1</sup> were 551 and 557 Fg<sup>-1</sup>, respectively. These values are about 2.7-fold higher than that of the pure MnO<sub>2</sub> film and about 1.4-fold higher than that of the Ag-doped MnO<sub>2</sub> film made without using surfactants.

**Keywords** Electrodeposition · Supercapacitor · Pseudocapacitor · Manganese oxide · Surfactant · Electrochemical capacitance

## Introduction

Global warming and shortages of fossil fuels encourage society to move towards sustainable and renewable energies with low carbon emissions [1]. There is increasing interest in renewable energy. However, the lack of good electrical energy storage systems remains a large problem [1]. In order to overcome this problem, high-efficiency energy storage devices such as batteries and electrochemical capacitors (ECs) are needed. ECs are referred to as supercapacitors because of their extraordinarily high capacitance density. They can physically store charges at their solid–liquid interfaces. This phenomenon is known as electrochemical double-layer capacitance (EDLC). Thus, ECs does not produce any volume change that usually accompanies redox reactions of the active masses in batteries. The charging and discharging rates of ECs are equally high while those of batteries are kinetically limited since they rely on reaction kinetics and mass transport [2].

An important subclass of EC devices consists of pseudocapacitors which undergoes both an electron transfer reaction and EDLC storage. Materials that exhibit pseudo-capacitive storage include conducting polymers [3] (i.e., polyaniline, polypyrrole, polythiophene, and poly(3,4-ethylenedioxythiophene)) and a variety of transition metal

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oxides. RuO<sub>2</sub> pseudocapacitors have relatively high specific capacitance (>1,000 Fg<sup>-1</sup>) in this class. However, their costs are prohibitive and their materials toxic [4]. Efforts to develop more practical and inexpensive pseudocapacitive materials such as MnO<sub>2</sub> [5–9] and NiO [10–12] are now quite active. MnO<sub>2</sub> is an important and well-studied material for supercapacitors. It exhibits theoretical specific capacitance of approximately 1,110 Fg<sup>-1</sup> through stoichiometric reduction of MnO<sub>2</sub> to MnOOH in a potential window of 1 V [13].

However, the capacitance for thick MnO<sub>2</sub> films (≈100 μm) is ultimately limited by the poor electrical conductivity of MnO<sub>2</sub>. Alternatively, the stability of ECs in the thin MnO<sub>2</sub> film (<5 μm) configuration is restricted because of low mass loading [14, 15]. In order to overcome the electrical resistance of MnO<sub>2</sub>, silver was incorporated into MnO<sub>2</sub> thin films. Ag mass loading of 50 μg cm<sup>-2</sup> was accomplished using cathodic electrodeposition. This led to higher specific capacitance and lower electrical resistance compared with pure MnO<sub>2</sub> films. Specific capacitance of 770 Fg<sup>-1</sup> was obtained for pure MnO<sub>2</sub> films at a scan rate of 2 mV s<sup>-1</sup> in 0.5 M Na<sub>2</sub>SO<sub>4</sub> electrolyte [6]. Previously, silver was incorporated in RuO<sub>2</sub> films using a radio frequency magnetron co-sputtering technique. The results showed that Ag-doped RuO<sub>2</sub> provided significantly higher specific capacitance than RuO<sub>2</sub> films. This was due to high electrical conductivity of Ag-doped RuO<sub>2</sub> [16].

In this work, we further improve the specific capacitance of Ag-doped MnO<sub>2</sub> films. This was done by controlling their morphology with surfactants in the electrodeposition process and increasing mass loading to an economical level (0.4–0.5 mg cm<sup>-2</sup>) [17, 18]. Use of surfactants for controlling morphology of the electrodeposited Ag-doped MnO<sub>2</sub> has not been done previously. However, the effect of surfactants on the morphology of the pure MnO<sub>2</sub> films was previously studied. Adding surfactants (i.e., CTAB and Triton X-100) to the electrodepositing solution of manganese sulfate can increase the surface area of MnO<sub>2</sub>. This leads to enhancement of the rechargeability of alkali manganese oxide batteries [19]. MnO<sub>2</sub> films electrodeposited in the presence of the Triton X-100 provide high porosity and therefore high surface area. This leads to 59% enhancement in specific capacitance when compared with values for MnO<sub>2</sub> film produced without Triton X-100 [5]. Structures of surfactants with nonpolar hydrocarbon chains and ionic or neutral polar head groups play an important role in electrosynthesis. This is because they can form micelle at solid–liquid interfaces [20–22]. Coulombic interactions between the head groups of surfactants and electrode surfaces are strong. Thus, micelles formed can be used as soft templates in the electrodeposition process [23].

## Experimental

### Chemicals and materials

Potassium permanganate (KMnO<sub>4</sub>), silver nitrate (AgNO<sub>3</sub>), sodium sulfate (Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>), sodium nitrate (NaNO<sub>3</sub>), potassium chloride (KCl), sodium dodecyl sulfate (SDS), cetyltrimethylammonium bromide (CTAB) purchased from the Sigma-Aldrich Co. were of analytical reagent grade. All solutions and subsequent dilutions were prepared using ultra-pure water (>18 MΩ cm). Graphite electrodes were supplied by Structural Probe, Inc.

### Electrodeposition

Electrodeposition experiments were conducted using a conventional three-electrode system. A 1 cm<sup>2</sup> graphite substrate, a platinum mesh, and an Ag/AgCl electrode (in saturated KCl) were used as the working, counter, and reference electrodes, respectively. Electrodeposition of MnO<sub>2</sub> was performed following a reported cathodic electrodeposition [6]. We further developed the process as follows. The electrodeposition was carried out at a current density of 1 mA cm<sup>-2</sup> vs. Ag/AgCl for 30 min in aqueous 20 mM KMnO<sub>4</sub> in 0.5 M NaNO<sub>3</sub> containing 0–1 mM AgNO<sub>3</sub> at 25 °C. The surfactants (CTAB or SDS) in concentrations of 0–1 mM were added to the process for the surface morphology-controlled samples. The as-electrodeposited films were annealed at 400 °C for 2 h to obtain a crystalline phase and to increase the purity of MnO<sub>2</sub> films [24].

### Structural and morphological characterizations

The microstructure and composition of the as-prepared films were investigated using scanning electron microscopy/energy dispersive X-ray spectroscopy (EDX) operating at 15 kV. The XRD patterns of samples were recorded using a diffractometer with CuK<sub>α</sub> radiation (λ=1.5406 Å) and a graphite monochromator at 40 kV and 30 mA.

### Electrochemical evaluation

Cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS) were used to evaluate the specific capacitance and stability of as-prepared electrodes. These techniques used a computer-controlled μ-AUTOLAB II potentiostat (Eco-Chemie, Utrecht, The Netherlands) equipped with a FRA2 frequency response analyzer module running GPES/FRA software. CV curves were scanned at voltage ramp rates of 1–100 mV s<sup>-1</sup>. EIS was carried out using a sinusoidal signal of 5 mV vs. Ag/AgCl over the frequency range from 100 kHz to 1 mHz. An aqueous electrolyte consisting of 0.5 M Na<sub>2</sub>SO<sub>4</sub> was used for all

capacitive measurements. This was done because when applied to supercapacitors, aqueous  $\text{Na}_2\text{SO}_4$  electrolyte, presents several advantages over the organic ones. It is relatively inexpensive, more environmentally friendly, and easier to make [25]. Additionally, the stability potential window of  $\text{Na}_2\text{SO}_4$  with regard to the pseudocapacitive mechanism is about twice as large as in acidic or basic electrolytes [25].

## Results and discussion

### Surface morphology

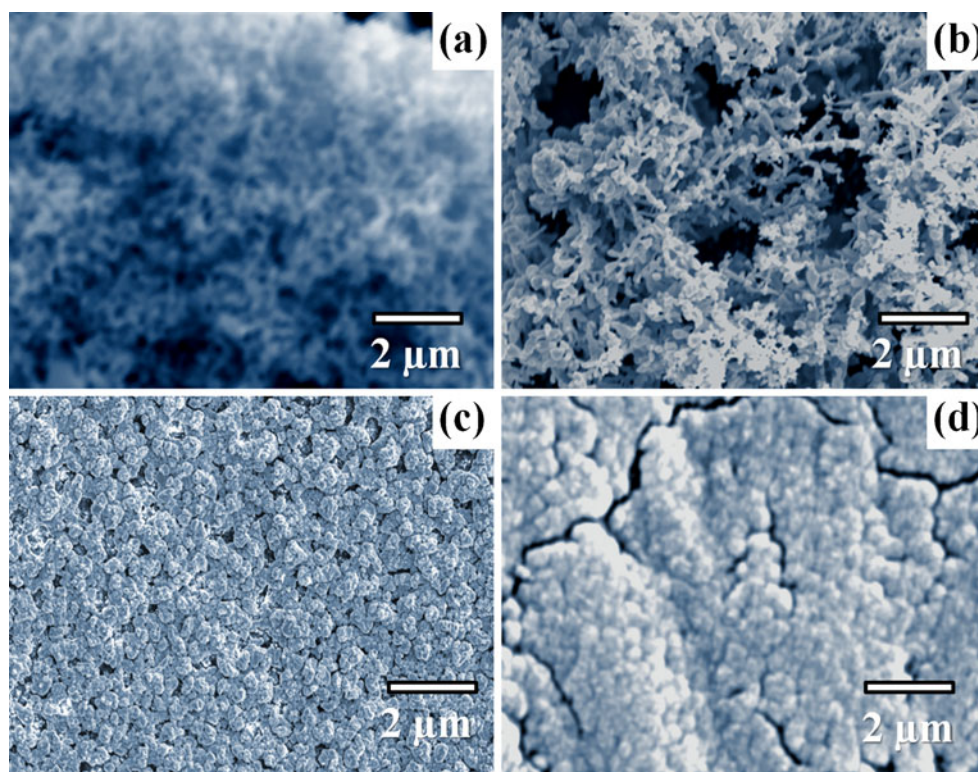
Scanning electron micrographs of the Ag-doped  $\text{MnO}_2$  films and the pure  $\text{MnO}_2$  film are shown in Fig. 1. In the presence of a cationic surfactant, CTAB, agglomerated oxide nanoparticles with a foam-like structure were observed (Fig. 1a). With an anionic surfactant, SDS, a dendrite Ag- $\text{MnO}_2$  structure is seen (Fig. 1b). In contrast, without structure-directing surfactants, denser film structures were observed on both Ag/ $\text{MnO}_2$  (Fig. 1c) and cracked  $\text{MnO}_2$  (Fig. 1d) films. The crack-free morphology of Ag-doped  $\text{MnO}_2$  is similar to materials recently reported [6]. This is possibly due to the strong interfacial interaction between the hybrid film and the graphite substrate. In contrast, the cracks typically found on the pure oxide films [26, 27] are because

of the huge stress generated by the capillary force during the drying process [28, 29]. Surfactants significantly affect the morphology of the Ag-doped  $\text{MnO}_2$  films. They play an important role in the nucleation and growth of hybrid films through the hemisphere-micelle formation at the electrode–liquid interface [19]. The structure of these products is controlled through a balance of interfacial forces between metal oxide precursor and surfactant [19, 30]. However, further studies on the nucleation mechanism for surfactant-assisted electrodeposition of foam-like and dendrite Ag-doped  $\text{MnO}_2$  structures are still needed.

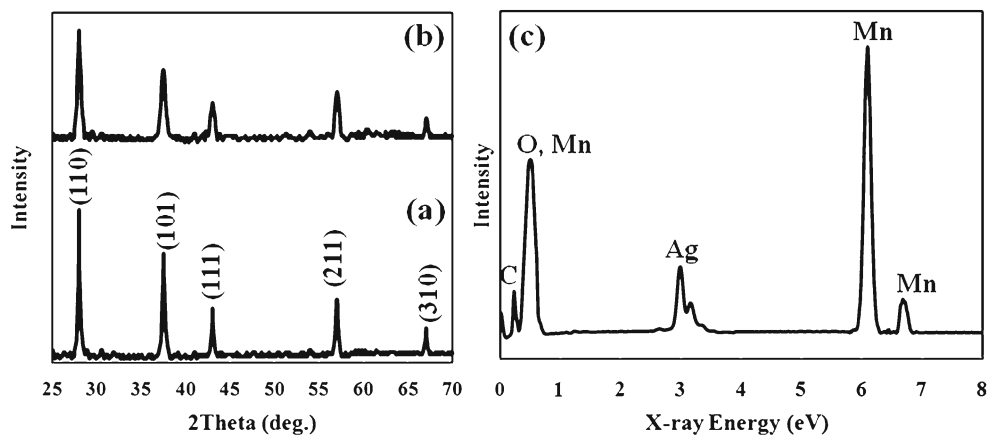
### Chemical structure of silver-doped manganese oxides

The phase and purity of the as-annealed oxide films were determined by powder XRD measurements. The XRD patterns in Fig. 2a, b show five dominant peaks. They can be indexed as (110), (101), (111), (211), and (310) planes of a pure tetragonal phase of  $\beta\text{-MnO}_2$  and are in good agreement with previous work [31, 32]. The XRD spectra also indicate that the films possess the high crystallinity of pure  $\beta\text{-MnO}_2$ . This is seen in the sharp and intense spectral peaks. Interestingly, in Fig. 2b, no peaks related to Ag are observed in Ag-doped  $\text{MnO}_2$  samples. This can be attributed to Ag ion insertion inside the tunnel of the  $\beta\text{-MnO}_2$  structure occurring more in doping than in the coating process. It is well established that  $\text{MnO}_2$  can accommodate large ions such as

**Fig. 1** Scanning electron micrographs of Ag-doped  $\text{MnO}_2$  films produced using **a** CTAB, **b** SDS, **c** without surfactant, and **d** the pure  $\text{MnO}_2$  film



**Fig. 2** X-ray diffraction patterns of **a** MnO<sub>2</sub> and **b** Ag-doped MnO<sub>2</sub> and **c** energy X-ray dispersive spectrum of Ag-doped MnO<sub>2</sub>



Na<sup>+</sup>, K<sup>+</sup>, Ag<sup>+</sup>, and Ca<sup>2+</sup> [33–35]. Typical EDX spectrum of the Ag-doped MnO<sub>2</sub> films in Fig. 2c shows Mn, Ag, O, and C elements. A peak at 0.64 keV is attributed to OL<sub>α1</sub>. Two peaks at 5.90 and 6.54 keV represent MnK<sub>α1</sub>, and MnK<sub>β1</sub>, respectively. The two predominant peaks at 3.00 and 3.16 keV result from AgL<sub>α1</sub> and AgL<sub>β1</sub>, respectively. A peak at 0.28 keV is found associated with CK<sub>α</sub> of the graphite electrode.

#### Electrochemical evaluation

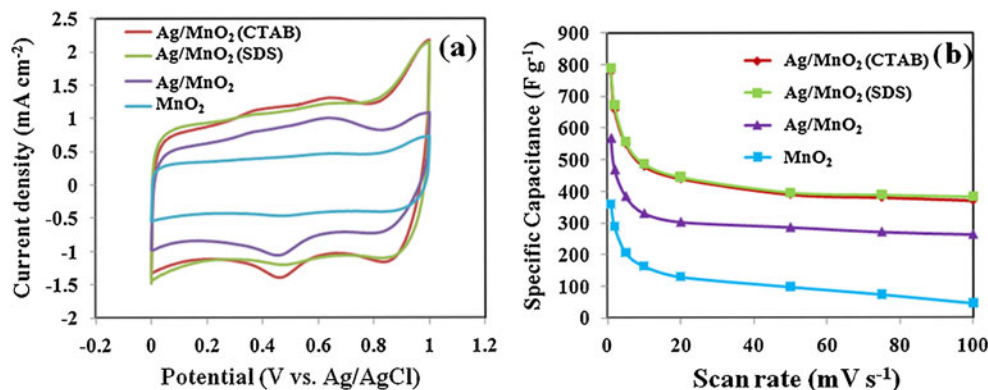
##### Cyclic voltammetry

The EDLC of the Ag-doped and pure MnO<sub>2</sub> films was investigated by means of the cyclic voltammetry in a 0.5 M Na<sub>2</sub>SO<sub>4</sub> electrolyte solution at potential intervals from 0 to 1 V vs. Ag/AgCl at the scan rate of 5 mV s<sup>-1</sup>. The voltammograms indicate pseudocapacitive behavior which stores energy through highly reversible surface redox (Faradic) reactions and the EDLC with highly reversible charging–discharging reactions of cations (i.e., Na<sup>+</sup> and H<sup>+</sup>) on the surface of oxide electrodes. The charge storage mechanism is attributed to the adsorption of Na<sup>+</sup> and H<sup>+</sup> on the surface of pseudocapacitor electrodes [14]. By integrating the

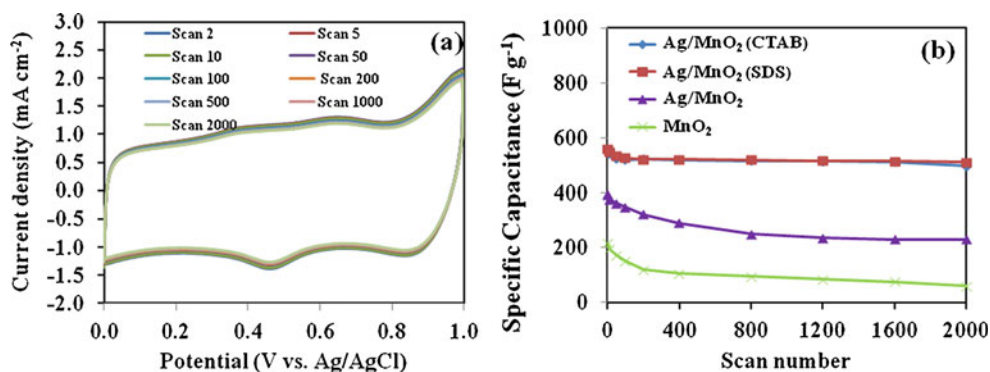
cathodic current for all potentials and subtracting the capacitive current contributed by the substrate, a value of the pseudocapacitive current ( $I_{cv}$ ) can be calculated. Using this value, along with the mass of deposition ( $m$ ) and the scan rate ( $v$ ), we can then calculate the specific capacitance ( $C$ ) according to the equation,  $C = I_{cv}/mv$ . From the CV in Fig. 3, the calculated  $C$  values of the Ag-doped MnO<sub>2</sub> electrodes produced using CTAB and SDS as soft templates, i.e., Ag/MnO<sub>2</sub> (CTAB) and Ag/MnO<sub>2</sub> (SDS) at the scan rate of 5 mV s<sup>-1</sup> were 551 and 557 F g<sup>-1</sup>, respectively. These values are about 1.4- and 2.7-fold higher than that of Ag/MnO<sub>2</sub> and MnO<sub>2</sub>, respectively.

CV measurements were also done at different scan rates over a range of 1–100 mV s<sup>-1</sup>. Calculated  $C$  values are shown in Fig. 3b. These values decrease with increasing scan rates since the scan rate makes a direct impact on the diffusion of Na<sup>+</sup> and H<sup>+</sup> into the pseudocapacitor electrodes. At high scan rates, the cations (Na<sup>+</sup> and H<sup>+</sup>) will primarily approach the outer surface of the electrode. At low scan rates, the cations can approach deep pores of the pseudocapacitor electrodes. Low scan rates allow more contribution to EDLC of the MnO<sub>2</sub>. However, a closer look at the slope of the specific capacitance curves of the Ag/MnO<sub>2</sub>(CTAB) and the Ag/MnO<sub>2</sub>(SDS) electrodes in Fig. 3b is revealing that the diffusion of

**Fig. 3** **a** Cyclic voltammograms of the pure MnO<sub>2</sub> and Ag-doped MnO<sub>2</sub> films in 0.5 M Na<sub>2</sub>SO<sub>4</sub> at a scan rate of 5 mV s<sup>-1</sup> and **b** specific capacitance as a function of the scan rate for which Ag-doped films were prepared from 20 mM KMnO<sub>4</sub> solution containing 1 mM AgNO<sub>3</sub> and 1 mM surfactants (i.e., SDS or CTAB)



**Fig. 4** **a** Cyclic voltammograms of Ag/MnO<sub>2</sub> (CTAB) at a range of scan (cycling) number (2–2,000) and **b** specific capacitance of all prepared films as a function of the scan number



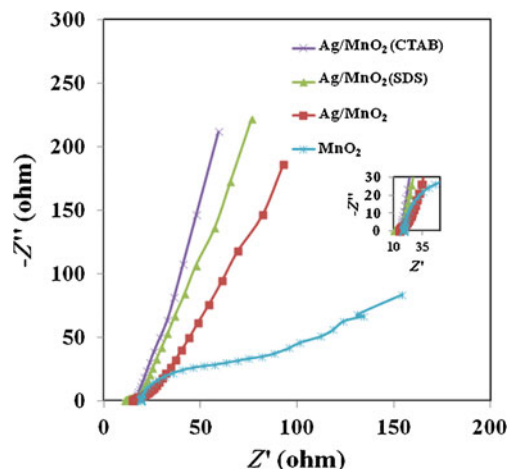
Na<sup>+</sup> and H<sup>+</sup> inside deep nanopores at high scan rates can significantly enhance pseudocapacitance. This suggests that the foam-like structure of Ag/MnO<sub>2</sub>(CTAB) and the dendrite structure of Ag/MnO<sub>2</sub>(SDS) electrodes offer a faster pathway for ion transport and lower resistance for charge and electrolyte diffusion.

The stability of doped pseudocapacitors has additionally been studied using the CV measurements over scan (cycling) number of 2,000. Figure 4a presents CV curves of Ag/MnO<sub>2</sub> (CTAB) upon cycling for 2,000 scans at the scan rate of 5 mV s<sup>-1</sup> with the cut-off voltages 0–1 V vs. Ag/AgCl in 0.5 M Na<sub>2</sub>SO<sub>4</sub>. The calculated value of C after 2,000 scans is only 1.4% less than that at scan number 2 (552 Fg<sup>-1</sup>). This indicates high Ag/MnO<sub>2</sub> (CTAB) stability. Figure 4b compares the C values of all prepared films as a function of the scan number. In this figure, Ag/MnO<sub>2</sub> (SDS) with its dendrite structure provides similar C values over the scan number of 2,000 with high stability. This is comparable to Ag/MnO<sub>2</sub> (CTAB). However, the Ag/MnO<sub>2</sub> electrode made without surfactants in the electrodeposition process shows less specific capacitance (about 1.5 times). The C values of Ag/MnO<sub>2</sub> still possess a significant two- to threefold increase over that of pure MnO<sub>2</sub>, which is in good agreement with very recent work [6].

*Electrochemical impedance spectroscopy*

The impedance plots in Fig. 5 show a partial semicircle at a high-frequency component. This is due to Faradic charge transfer resistance and a nearly straight line along the imaginary axis at a low-frequency component due to the mass transport limit. The smaller semicircles (inset of Fig. 5) in the high-frequency range are observed in the case of Ag-doped MnO<sub>2</sub> films. This can be compared with the larger semicircle of pure MnO<sub>2</sub>. From this, it can be suggested that incorporation of silver into MnO<sub>2</sub> structure leads to fast electron transfer across electrified interfaces or an inner Helmholtz layer to the solvated cations in the outer Helmholtz layer. This process

enhances the specific capacitance of pseudocapacitor electrodes as evidenced from the cyclic voltammetry. Additionally, the magnitude of charge-transfer resistances at the solid–liquid interface is in the following order: Ag/MnO<sub>2</sub>(SDS) ≈ Ag/MnO<sub>2</sub>(CTAB) < Ag/MnO<sub>2</sub> < MnO<sub>2</sub>. If charge-transfer resistance is smaller, the pseudocapacitor provides higher specific capacitance values [10]. The slope of the straight line in a low frequency range is due to the Warburg resistance. This is a result of the frequency dependence of ion diffusion from the electrolyte to the electrode interface. From the Nyquist curves, the slopes of the curves for the Ag/MnO<sub>2</sub> (SDS) and Ag/MnO<sub>2</sub> (CTAB) electrodes are steeper than the slope for Ag/MnO<sub>2</sub> and the pure MnO<sub>2</sub>. This implies that the dendrite and foam-like structures of Ag/MnO<sub>2</sub> can significantly enhance diffusion of solvated cations to the porous electrodes. If we compared the Nyquist curves of Ag/MnO<sub>2</sub> and the pure MnO<sub>2</sub>, doping silver into MnO<sub>2</sub> structure enhanced the electronic conductance of the oxide supercapacitor as well as the ion transport [36, 37].



**Fig. 5** Nyquist plots of as-prepared electrodes in 0.5 M Na<sub>2</sub>SO<sub>4</sub> using a sinusoidal signal of 5 mV over the frequency range from 100 kHz to 1 mHz

## Conclusions

In summary, MnO<sub>2</sub> and Ag-doped MnO<sub>2</sub> pseudocapacitors at a mass-loading level of 0.4 mg cm<sup>-2</sup> were coated on graphite electrodes by using an electrodeposition. With the presence of surfactants (i.e., SDS or CTAB) in the electroplating solution, dendrite and foam-like structures of Ag-doped MnO<sub>2</sub> were successfully produced in lieu of denser films produced without surfactants in the solution. The specific capacitance values of the Ag/MnO<sub>2</sub> (CTAB) and Ag/MnO<sub>2</sub> (SDS) in 0.5 M Na<sub>2</sub>SO<sub>4</sub> at the scan rate of 5 mV s<sup>-1</sup> were 551 and 557 F g<sup>-1</sup>, respectively. These values are about 2.7-fold higher than that of the pure MnO<sub>2</sub> and about 1.4-fold higher than that of the Ag-doped MnO<sub>2</sub> film without using surfactants. The foam-like structure of Ag/MnO<sub>2</sub> (CTAB) and the dendrite structure of Ag/MnO<sub>2</sub> (SDS) electrodes with high stability play an important role in the specific capacitance of pseudocapacitors by enhancing ion transport and lowering resistance for charge and electrolyte diffusion. These Ag-doped MnO<sub>2</sub> pseudocapacitor electrodes with foam-like and dendrite structures providing high specific capacitance and durability may be used for practical applications.

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